

INCOMMENSURATE STRUCTURES OF  $\text{OsF}_6$  AND  $\text{MoF}_6$  GRAPHITE INTERCALATES  
STUDIED BY NEUTRON SCATTERING

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ABSTRACT

Neutron scattering has been used to study the in-plane structure of stage-1 graphite intercalated  $\text{C}_{8.4}\text{MoF}_6$  and  $\text{C}_9\text{OsF}_6$ . Both compounds exhibit a first order phase transition at 280K from a liquid to a modulated 2D incommensurate structure at low temperatures. The structure can be described as a domain-wall modulated  $(2 \times 2)\text{Ra}^\alpha$  where  $\alpha$  is a small rotation angle. In  $\text{C}_{8.4}\text{MoF}_6$  the domain-wall lattice is correlated over  $250\text{\AA}$  which is close to the instrumental resolution, whereas for the  $\text{C}_9\text{OsF}_6$  the correlation range is only  $60\text{\AA}$ . The relation between the residual resistivity and the intercalant structure is discussed.

INTRODUCTION

Graphite intercalation compounds with metal hexafluorides, HOPG/ $\text{MoF}_6$  and HOPG/ $\text{OsF}_6$ , have been extensively studied in the past several years using ESR [1], susceptibility [2], reflectivity [3] and in-plane and c-axis resistivities [4]. It was found that the charge transfer per intercalant species is one electron for stage I  $\text{C}_9\text{OsF}_6$  but only 0.2 electrons for stage I  $\text{C}_{8.4}\text{MoF}_6$  [1,2]. The charge transfer per carbon atom, however, is only slightly larger for the former GIC, suggesting possible charge localization [3]. The in-plane residual resistivity for  $\text{C}_9\text{OsF}_6$  (stage I) was found to be larger than that of  $\text{C}_{8.4}\text{MoF}_6$  by several orders of magnitude.

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Here we present neutron diffraction and in-plane resistivity data, for stage I  $C_{9.0}OsF_6$  and  $C_{8.4}MoF_6$ . The in-plane structure is modulated by domains and domain walls due to strong coupling with the graphite lattice. We demonstrate a correlation between the intercalant domain size and the residual resistivities.

## EXPERIMENTAL DETAILS AND RESULTS

### Sample Preparation

Samples were prepared from HOPG and characterized by methods described previously [1,2].

### Neutron diffraction

The neutron diffraction experiments were carried out using a triple axis spectrometer situated at a cold source beam at the Riso Dr-3 reactor. For the

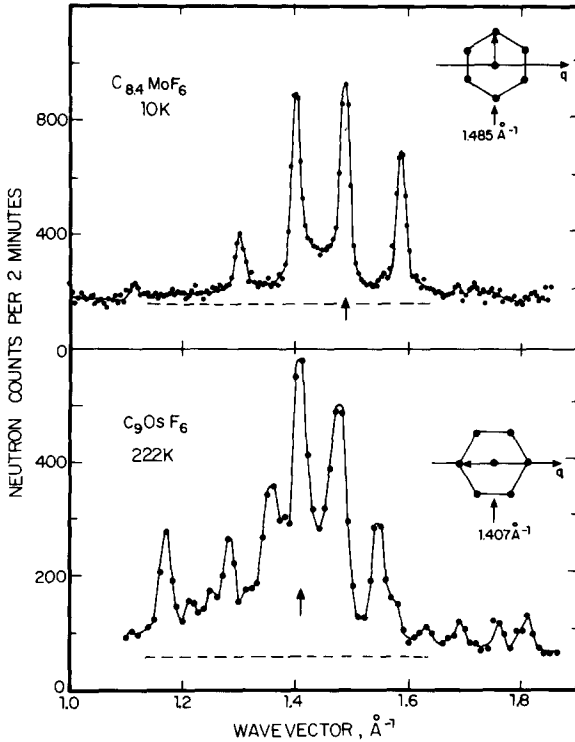


FIG. 1. High resolution neutron diffraction scans with momentum transfer in the plane for  $C_{8.4}MoF_6$  and  $C_{9.0}OsF_6$ . The full line is a guide to the eye and the broken line indicates the level of background scattering. Note that for  $C_{9.0}OsF_6$  the spectrum is temperature independent from 250K down to 4K. The inserts illustrate the satellite patterns near the  $\tau_T(10)$  reflections which give a satisfactory assignment of the observed peaks. The arrows indicate the positions for the fundamental peaks of the intercalant structure.

initial scans a wavelength of  $2.38\text{\AA}$  was used, at which pyrolytic graphite filters are very efficient in removing any higher order contamination. High resolution scans were performed using wavelength of  $4\text{\AA}$  and  $4.3\text{\AA}$ . A 10-cm cooled Be-filter placed before the graphite (002) monochromator and, in the latter case, combined with a 5 cm graphite filter, was used to remove the higher orders. The typical collimation was  $60'$ , but occasionally  $30'$  was used before the monochromator and after the sample. The best resolution achieved was  $0.0095\text{\AA}^{-1}$ , full width at half maximum (FWHM). The samples were mounted in standard sealed aluminum containers, filled with He in a dry box. The c-axis was in the scattering plane and the cooling was achieved by means of a closed cycle refrigerator.

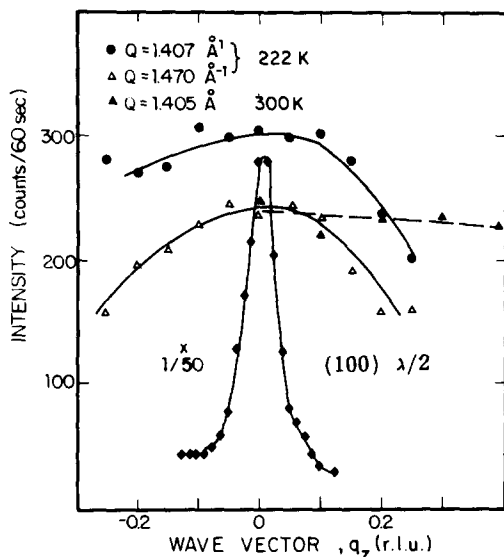


FIG. 2. Scans along  $q_z$  for the most prominent in-plane peaks for  $C_9OsF_6$  together with a similar scan for the  $\tau(100)\lambda/2$  reflection. The weak  $q_z$  dependence corresponds to 2D nature of the intercalant structure.

The staging index and the A-A stacking of the carbon layers were confirmed by measurements of several of the fundamental graphite reflections. The intercalant structure is seen most clearly in the in-plane scans and Fig. 1 shows examples of such scans in the low temperature phases of the two compounds. The most intense scattering is found in the wavevector region  $1\text{\AA}^{-1} < q < 2\text{\AA}^{-1}$ . The region  $2\text{\AA}^{-1} < q < 3\text{\AA}^{-1}$  contained very little intercalant scattering in accordance with calculations of the average form factor of the  $MF_6^-$  molecules. Weak intercalant peaks were observed in the range  $3\text{\AA}^{-1} < q < 4\text{\AA}^{-1}$ , but we were unable to assign those. Scans along  $q_z$  for the most prominent in-plane intercalant peaks for  $C_9OsF_6$  are shown in Fig. 2 together with similar scans of the  $\tau_G(100)$ ,  $\lambda/2$  whose width in  $q_z$  is determined by the mosaic of the sample. Clearly the

intercalant peaks are much broader, which is a signature of the 2-D nature of the low temperature structure of the compounds. Similar results were obtained for HOPG/MoF<sub>6</sub>. The intercalant structures become disordered above 280K and the temperature evolution of the diffraction patterns and of the c-axis resistivity was followed in detail through the transition region from 30K upwards. Both types of measurements show hysteresis and, furthermore, the low temperature diffraction patterns were found to be dependent on the thermal history. However, slow cooling (<0.5K/min.) from the fully disordered state gave reproducible results.

### Resistivity Measurements

The in-plane resistivity was measured by a contactless method using a ferrite core adapted to a helium flux system [4]. Figs. 3 present the in-plane resistivity as a function of temperature for some of the compounds. The resistivity data were fitted to the relation  $\rho_a = A + BT + CT^2$  and the various parameters are given in Table 1 for some of the samples.

TABLE I

	A ( $\mu\Omega$ cm)	B $\mu\Omega$ cm/K	C $\mu\Omega$ cm/K <sup>2</sup>
C <sub>9</sub> OsF <sub>6</sub> (I)	100	---	---
C <sub>12</sub> OsF <sub>6</sub> (I)	28.0	$2.8 \times 10^{-2}$	$2.5 \times 10^{-4}$
C <sub>8</sub> MoF <sub>6</sub> (I)	0.6	$5 \times 10^{-4}$	$5 \times 10^{-5}$
C <sub>10</sub> MoF <sub>6</sub> (I)	1.95	$2 \times 10^{-3}$	$5 \times 10^{-5}$

The most striking feature is the significantly larger residual resistivity for C<sub>9</sub>OsF<sub>6</sub> as compared to C<sub>9</sub>MoF<sub>6</sub>. Using a Drude type model we have estimated a mean free path length, L, of  $L = 20\text{\AA}$  for C<sub>9</sub>OsF<sub>6</sub> and  $L = 3000\text{\AA}$  for C<sub>9</sub>MoF<sub>6</sub>. The tunneling model of Ulloa and Kirczenow [5] gives a domain size of  $L(\text{DH}) = 20\text{\AA}$  for C<sub>9</sub>OsF<sub>6</sub> and  $L(\text{DH}) = 3020\text{\AA}$  assuming a wall thickness of  $8\text{\AA}$ .

### Structural Model

The observed diffraction patterns at low temperatures can be interpreted in terms of a model based on a close-packed 2-D intercalant lattice (neglecting molecular orientation effects) with a lattice parameter,  $a_I$ , near that of the (2x2)-registered structure. For C<sub>9</sub>OsF<sub>6</sub> the lattice mismatch, ( $a_G/2a_I$ ) is 0.950, whereas it is very close to 1.0 for C<sub>8.4</sub>MoF<sub>6</sub>. The harmonic interaction between the graphite and the intercalant lattices induce a modulation pattern with characteristic Fourier-components given by  $\tau_s = \tau_G - \tau_I$ , where  $\tau_G$  denotes a reciprocal lattice vector associated with the graphite layers, and  $\tau_I$  denotes the corresponding vectors of the intercalant lattice [6]. The modulation pattern gives rise to a decoration of the fundamental reflection of both lattices with a

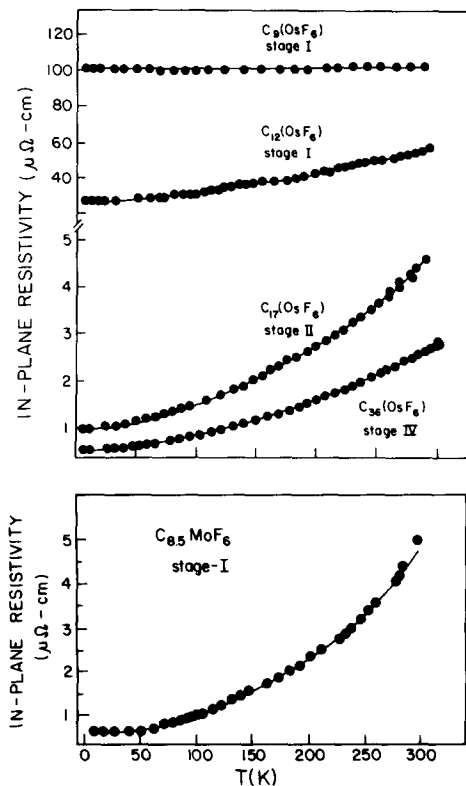


FIG. 3. The temperature dependence of the in-plane conductivities for different stages C/OsF<sub>6</sub> and for stage I C<sub>8.5</sub>MoF<sub>6</sub>

hexagonal pattern of satellite reflections. The satellite intensities reflect the actual distortions. In the  $q$ -range shown in Fig. 1 we observe mainly the satellites near the  $\tau_I(10)$ -reflection in a projection onto the  $q$ -axis, due to the powder average in the plane. The simplest assignment of these peaks can be made on the basis of the satellite patterns shown schematically as inserts in Fig. 1. With this assignment we deduce that the MoF<sub>6</sub>-intercalant lattice is very near to  $(2 \times 2)$ -registry and that the small mismatch in the lattice parameters gives rise to a rotational epitaxy dominated by transverse modulations. The modulation vector is  $|\tau_s| = 0.106 \text{\AA}^{-1}$  which corresponds to a relative rotation of the graphite and the intercalant lattices of  $\alpha = 2.8^\circ$ . The larger lattice parameter mismatch for the OsF<sub>6</sub>-compound results in the modulation pattern of predominantly longitudinal character where the length of the modulation vector  $|\tau_s| = 0.125 \text{\AA}^{-1}$  is close to the lattice mismatch. In both cases the modulation pattern can be thought of as a domain-wall lattice with domains of linear dimensions  $1/|\tau_s|$  corresponding to 20-25  $a_G$ -units. The domains are separated by fairly sharp domain walls as can be deduced from the facts that (1) the satellites are as intense as the fundamental reflections, and (2) satellites up to

high orders are observed. However, a more quantitative analysis will require accurate single crystal structure factor data. A closer examination of the peak positions shows that they deviate systematically from the simple assignment based on the harmonic model, especially for the  $\text{OsF}_6$ -compound. We interpret this as evidence for distortions of the domain-wall lattices, probably due to higher-order registry effects. In Table 2 we list the calculated peak positions

TABLE 2

Comparison of the observed and calculated peak positions for the domain-wall lattices in  $\text{C}_9\text{OsF}_6$  and  $\text{C}_{8.4}\text{MoF}_6$ . Model (1) is hexagonal and models (2) and (3) have orthorhombic distortions. For further details see Ref. [6]. All the values are in  $\text{\AA}^{-1}$  units.

	Observed positions	model 1	model 2	model 3
$\text{C}_9\text{OsF}_6$ $\tau_s = 0.125$ $\tau_I = 1.407$	1.169	1.169	1.1701	1.170
	1.277	1.289	1.284	1.281
	1.360	1.352	1.369 + 1.337	1.340 + 1.353
	1.407	1.4085	1.4085	1.405
	1.470	1.472	1.45 + 1.487	1.463 + 1.478
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$\text{C}_{8.4}\text{MoF}_6$ $\tau_s = 0.106$ $\tau_I = 1.485$	1.112	1.118		
	1.30	1.301		
	1.398	1.394		
	1.485	1.485		
	1.583	1.573		
	1.684	1.667		

for several models for the distorted lattices. In addition to the small shifts we also observe an intrinsic broadening of the diffraction lines, especially for the  $\text{OsF}_6$ -compound. This indicates that the domain-wall lattices have a finite correlation length, probably due to pinning effects. For  $\text{C}_9\text{OsF}_6$  we find a correlation range of  $60\text{\AA}$ , whereas the range is close to the instrumental resolution of  $250\text{\AA}$  for the  $\text{MoF}_6$ -compound.

## SUMMARY

It is tempting to associate the in-plane structure of the intercalant as measured by neutron scattering with the in-plane resistivity (note that the latter measure the domain size on the graphite layers). Our results strongly suggest smaller domain size for  $\text{C}_9\text{OsF}_6$  from both resistivity and neutron scattering. However, the size of the domains are different in both measurements. We believe that the small intercalant domains in the case of  $\text{C}_9\text{OsF}_6$  induce defects on the graphite layer. This effect and the charge localization found in this compound are probably responsible for the large residual resistivity of  $\text{C}_9\text{OsF}_6$ .

Summarizing, we find that the MF<sub>6</sub> GIC constitute a very interesting family of materials with intriguing chemical, electronic and structural properties and, clearly, considerably more study is needed before we can claim to understand these properties.

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